

# Materials and devices design for efficient double junction polymer solar cells

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## ABSTRACT

Organic solar cells exhibit potential to provide light-weight and low-cost solar energy on flexible substrates. However, current efficiency is still low for applications. New materials and device designs are needed to increase cell efficiency and make this technology available for large-scale applications. The dependence of double junction solar cell efficiency on polymer bandgaps in top and bottom subcells are presented, which provides guidance for engineering new conjugated polymers for efficient photovoltaic device development. The achievable cell efficiency can be beyond 16% with the bandgap of the bottom subcell at  $\sim 1.6$  eV ( $\sim 775$  nm) and that of the top subcell at  $\sim 1$  eV ( $\sim 1240$  nm). In addition, the LUMO and HOMO energy levels of the donor polymers are provided depending on various acceptor materials such as PCBM, TiO<sub>2</sub>, ZnO and CdSe. The interfacial layers between the subcells in double junction organic devices are also discussed.

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## 1. Introduction

Alternative energy has attracted increasing interest to mitigate dependence on conventional fossil fuels, with sunlight emerging as a particularly promising clean and readily available source. There are a number of reasons we have not yet effectively harnessed the power of the sun. Silicon (Si) solar cells usually require complex high-temperature and vacuum processing, and high purity silicon, making them cost prohibitive as an energy source [1–3]. These cells are further hampered by limited mechanical flexibility. Polymer solar cells exhibit potential as an inexpensive alternative to Si solar cells due to their solution-based processing [4,5]. Conjugated polymers also offer an attractive approach for increasing solar cell efficiencies because their bandgaps and energy levels can be engineered by modifying their chemical structure. However, efficiency of polymer solar cells does not yet approach that of inorganic solar cells. Low carrier mobilities and short carrier diffusion lengths in existing conjugated polymers make it impossible to arbitrarily increase active layer thickness in order to allow full spectrum light absorption in single junction solar cells. One promising path to increase cell efficiency is to use varied bandgaps in a serial structure in which two or more subcells with complementary absorption spectra are stacked [6–25]. In a two-terminal multijunction cell, open circuit voltage ( $V_{oc}$ ) is the sum of  $V_{oc}$ 's of individual subcells, while

current is determined by the minimum one in individual subcells [2,6,8,14–22,26].

Cell efficiencies have reached beyond 8% for single junction polymer solar cells, about 10% for double junction polymer cells. However, further significant increase in cell efficiency requires new design in materials and devices. Polymers currently lack appropriate bandgaps and HOMO/LUMO energy levels for optimal operation with highest efficiency in single and double junction polymer solar cells. There is a critical need for research in synthesizing and understanding a plethora of novel variable bandgap polymers with high carrier mobilities and controllable HOMO/LUMO energy levels to construct high efficiency polymer solar cells. Advances in efficiency in double junction cells also require greater fundamental understanding of interfacial layers between subcells. This work provides a guide for researchers to design and synthesize new polymers with a clear target in terms of polymer bandgaps, LUMO and HOMO energy levels, as well as selection of proper interfacial layers between subcells in a double junction device structure.

## 2. Results and discussion

Previous reports by Siddiki et al. [2] and Scharber et al. [27] showed that single junction polymer solar cells could achieve efficiencies up to 10–13%. Further increase in efficiency of single junction cells is challenging partly due to thermalization loss caused during the conversion from a large energy photon into an electron–hole pair in low bandgap polymers. Fortunately borrowed from the concept of inorganic solar cells [28], double

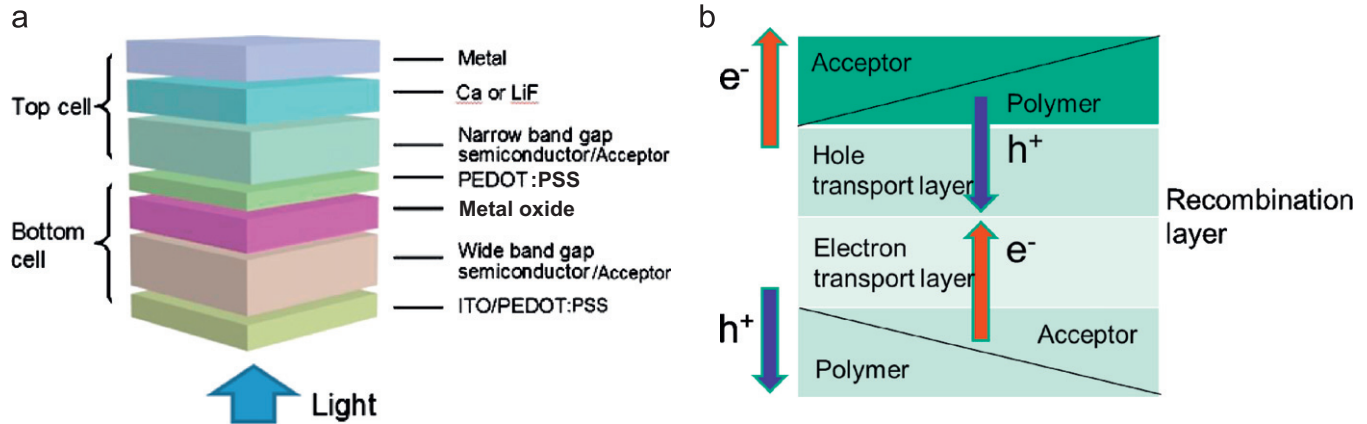
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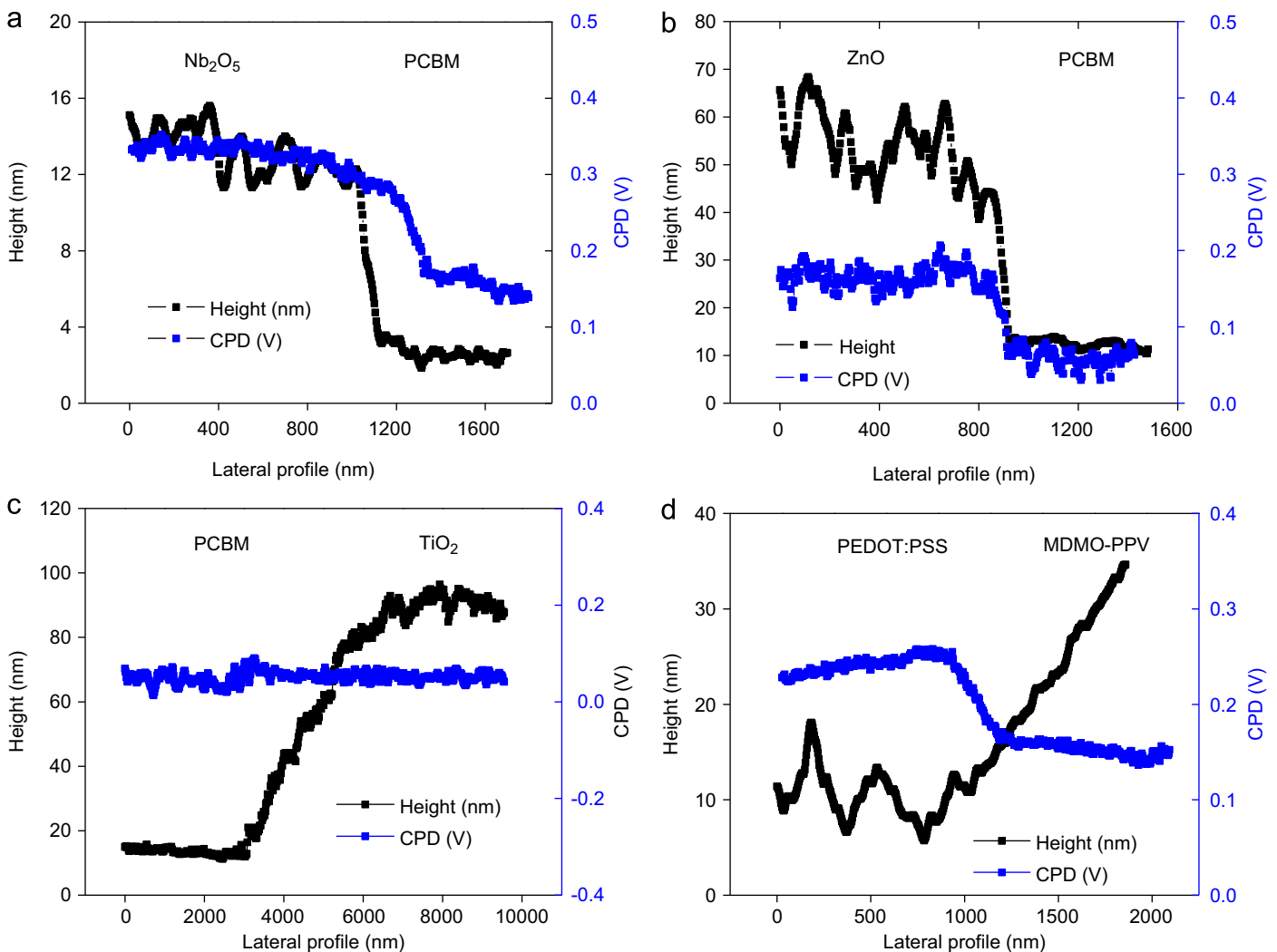
junction solar cell structures appear to be a promising solution to achieve a higher efficiency in polymer solar cells.

In series connection double junction solar cells, the overall current is limited by the lowest current produced from individual subcells. To maximize current output from double junction cells, the currents need to be matched between individual junctions.

Current matching can also help prevent the build-up of photo-generated charges in local regions in the cells [2,6,29]. The built-up charges can lead to formation of local potential and electrical field that affect performance of double junction solar cells by deviating from the optimal power output point, thereby reducing  $V_{oc}$ , short circuit current density ( $J_{sc}$ ), or fill factor ( $FF$ )



**Fig. 1.** (a) Device configuration of a typical double junction polymer solar cell and (b) the mechanism of the interfacial layers consisting of HTL and ETL. The HTL is typically PEDOT:PSS, while ETL is  $\text{TiO}_2$ , ZnO or  $\text{Nb}_2\text{O}_5$ .



**Fig. 2.** Height and surface potential across a surface interface measured by scanning probe microscope (SPM) of (a)  $\text{Nb}_2\text{O}_5$  on top of PCBM, (b) ZnO on top of PCBM, (c)  $\text{TiO}_2$  on top of PCBM, and (d) MDMO-PPV on top of PEDOT:PSS.

[2]. Current matching can be achieved by selecting appropriate donors, varying active layer thicknesses, and/or engineering donor–acceptor morphology and compositions. The overall double junction solar cell current density ( $J_{sc-double}$ ) in series connection is equal to the smallest one from its corresponding individual subcell. The overall open circuit voltage ( $V_{oc-double}$ ) of double junction solar cells in series connection is the sum of  $V_{oc}$ 's of individual subcells as  $V_{oc-double} = V_{oc1} + V_{oc2}$ . The  $V_{oc1}$  and  $V_{oc2}$  are the open circuit voltage of respective individual subcells.

Deposition of double or multiple subcells in series can lead to formation of inverse junctions between adjacent subcells. Such inverse junctions can behave as potential barriers for charge flow from one subcell to the next. To solve this issue, interfacial layers including electron transport layer (ETL) and hole transport layer (HTL) are required to be inserted between adjacent subcells as a tunneling/recombination layer. This can also help align Fermi levels of adjacent subcells and reduce energy loss in the tunneling/recombination process, thereby minimizing the photovoltage loss of the overall double junction cells [29]. In addition to the above mentioned purpose of tunneling/recombination, the interfacial layers play two more roles: (1) a protective layer for the bottom subcell and (2) a foundation for the top subcell [2]. The interfacial layers are required to be transparent or at least semi-transparent to low energy photons because non-transparent interfacial layers block the light from arriving at the second subcell. Transparent metal oxides such as  $TiO_{x(x \leq 2)}$  and ZnO were reported to serve as ETL, while poly(3,4-ethylenedioxythiophene) doped with poly(styrenesulfonate) (PEDOT:PSS) and other transitional metal oxides (e.g.  $V_2O_5$  and  $MoO_3$ ) act as HTL [2,25,30]. The interfacial layers can be prepared by spin coating (e.g. ZnO and  $Nb_2O_5$ ) [11,31], dip coating (e.g.  $TiO_2$ ), dc magnetron sputtering (e.g. ITO) [9], and thermal evaporation (e.g. Sn, Au, Ag,  $WO_3$ ) [6,11].

Fig. 1a shows device configuration of a typical double junction polymer solar cell. The bottom subcell usually uses wide bandgap polymers, while the top utilizes narrow bandgap polymers. As mentioned above, the interfacial layers typically consist of HTL

such as PEDOT:PSS and ETL such as  $TiO_2$ , ZnO and  $Nb_2O_5$ . Fig. 1b shows the operation mechanism of the interfacial layers consisting of HTL and ETL. The electrons from the top subcell transport to the overall top electrode, while the holes from the bottom one transport to the overall bottom electrode, forming current flow in the external circuit. In the meanwhile, the holes from the top subcell transfer to the HTL, while the electrons from the bottom transfer to the ETL. Such holes and electrons transport through the respective transport layer and then tunnel/recombine at the interfacial layers.

The surface potentials (or contact potential difference (CPD)) of  $Nb_2O_5/PCBM$ , ZnO/PCBM,  $TiO_2/PCBM$  and MDMO-PPV/PEDOT:PSS were studied using Agilent SPM 5500. As shown in Fig. 2, the height plots confirmed that  $Nb_2O_5$ , ZnO, and  $TiO_2$  were deposited on top of the PCBM layer, while MDMO-PPV was deposited onto PEDOT:PSS. The surface potential (or contact potential difference (CPD)) results showed that an energy barrier of 0.25 eV existed for electron transfer from PCBM to  $Nb_2O_5$  (Fig. 2a), while such barrier was only 0.12 eV from PCBM to ZnO (Fig. 2b). However, a very small energy barrier was observed for electron transfer from PCBM to  $TiO_2$  (Fig. 2c). In addition, we also studied the surface potential on the PEDOT:PSS/MDMO-PPV samples and the results showed that hole transfer from MDMO-PPV to PEDOT:PSS was energetically favorable (Fig. 2d).

Fig. 3 shows the dependence of calculated overall double junction cell efficiency on the bandgaps of the polymers in the bottom and top subcells. It can be seen that the bandgaps of the bottom and top subcells need to be carefully selected within the inner regions in Fig. 3 to achieve the highest possible efficiency for double junction polymer solar cells. Fig. 4 shows the calculated photovoltaic performance of double junction polymer solar cells connected in series in terms of  $V_{oc}$ ,  $J_{sc}$ , and cell efficiency ( $\eta$ ) with a dependence on the absorption cutoff wavelength of the bottom and top subcells. The calculation was conducted with following presumptions: (1) individual subcells are connected in series, (2) each subcell absorbs equally divided amount of photon flux and generates identical current density,

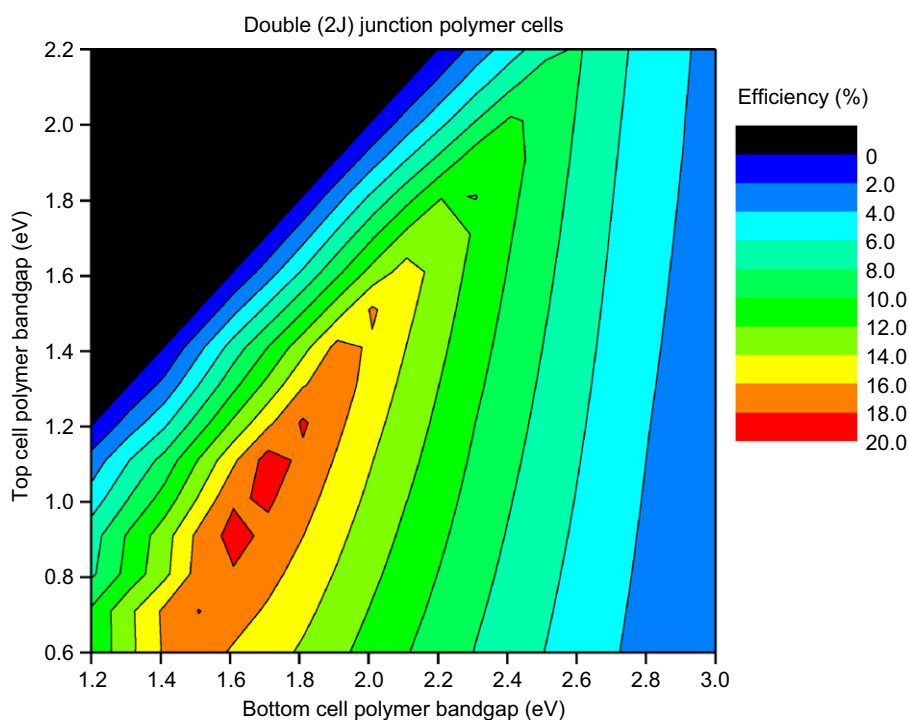
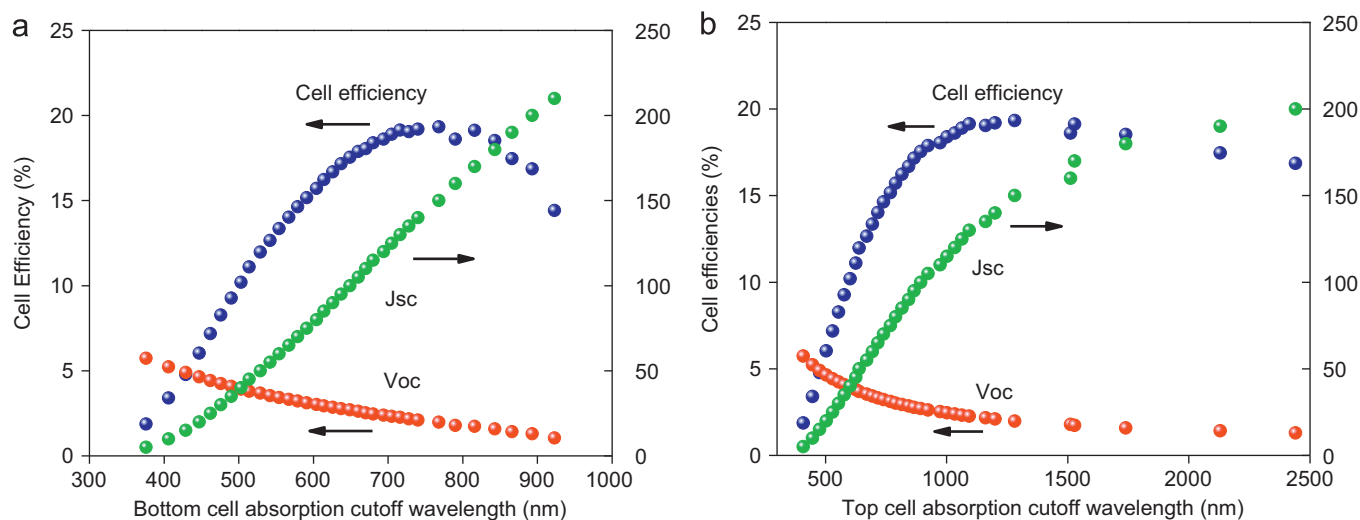


Fig. 3. The dependence of calculated overall double junction cell efficiency on polymer bandgaps in the bottom and top subcells.



**Fig. 4.** The calculated photovoltaic performance of double junction polymer solar cells connected in series in terms of  $V_{oc}$ ,  $J_{sc}$ , and  $\eta$  with a dependence on the absorption cutoff wavelength of the (a) bottom and (b) top subcell.

**Table 1**  
Polymer parameters (bandgaps, LUMOs and HOMOs) for efficient double junction polymer solar cells based on acceptors such as PCBM, TiO<sub>2</sub>, ZnO and CdSe. The LUMOs of PCBM and CBs of TiO<sub>2</sub>, ZnO, and CdSe were adopted from Ref. [2]. It is presumed that an offset of 0.3 eV is needed for exciton dissociation.

Acceptor	Top subcell (eV)			Bottom subcell (eV)		
	Polymer bandgap	Polymer LUMO	Polymer HOMO	Polymer Bandgap	Polymer LUMO	Polymer HOMO
PCBM LUMO = -4.3 eV	~1	-4.0	-5.0	~1.6	-4.0	-5.6
TiO <sub>2</sub> CB = -4.4 eV	~1	-4.1	-5.1	~1.6	-4.1	-5.7
ZnO CB = -4.2 eV	~1	-3.9	-4.9	~1.6	-3.9	-5.5
CdSe CB = -3.8 eV	~1	-3.5	-4.5	~1.6	-3.5	-5.1

leading to minimal current loss, (3) the bandgap ( $E_g$ ) of the donor in individual subcells is obtained from their cutoff absorption wavelengths and the  $V_{oc}$  of individual subcells was estimated by subtracting  $E_g$  from exciton binding energy ( $E_b$ ), and (4) the  $FF$  and incident photon to current efficiency ( $IPCE$ ) were 0.65 and 0.60, respectively. The cell efficiencies of double junction solar cells were finally calculated using the equation

$$\eta_{double} = \frac{J_{sc-double} V_{oc-double} FF}{P_{light}}$$

As the light absorption spectrum is extended into longer wavelength (red or near infrared) regions, the polymer bandgaps decrease, which increases  $J_{sc}$  but reduces  $V_{oc}$ , as shown in Fig. 4. Therefore, a balance between  $J_{sc}$  and  $V_{oc}$  needs to be obtained so that their product along with  $FF$  ( $J_{sc} \times V_{oc} \times FF$ ) is maximized. An optimal light absorption spectrum range of both the bottom and top subcells needs to be identified to achieve the highest possible efficiency. In order to maximize the overall cell efficiency, the bandgap of the bottom subcell would be at ~1.6 eV (~775 nm), while that of the top subcell at ~1 eV (~1240 nm). In addition to the bandgaps in the subcells, the absolute energy levels including LUMO and HOMO are also critical. Table 1 presents the LUMO and HOMO energy levels of donor polymers combined with various acceptor materials such as PCBM, TiO<sub>2</sub>, ZnO and CdSe.

### 3. Conclusions

The materials and devices design for efficient double junction polymer solar cells are presented. An optimal light absorption spectrum range of both the bottom and top subcells in double junction cells needs to be identified to achieve the highest

possible energy conversion efficiency. The achievable cell efficiency can be beyond 16% with the bandgap of the bottom subcell at ~1.6 eV (~775 nm) and that of the top subcell at ~1 eV (~1240 nm). In addition, the LUMO and HOMO energy levels of the donor polymers are provided depending on various acceptor materials such as PCBM, TiO<sub>2</sub>, ZnO and CdSe. The interfacial layers between the subcells in the double junction organic devices are also discussed. This work may provide guidance to design and fabricate high efficiency single and double junction polymer solar cells.

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