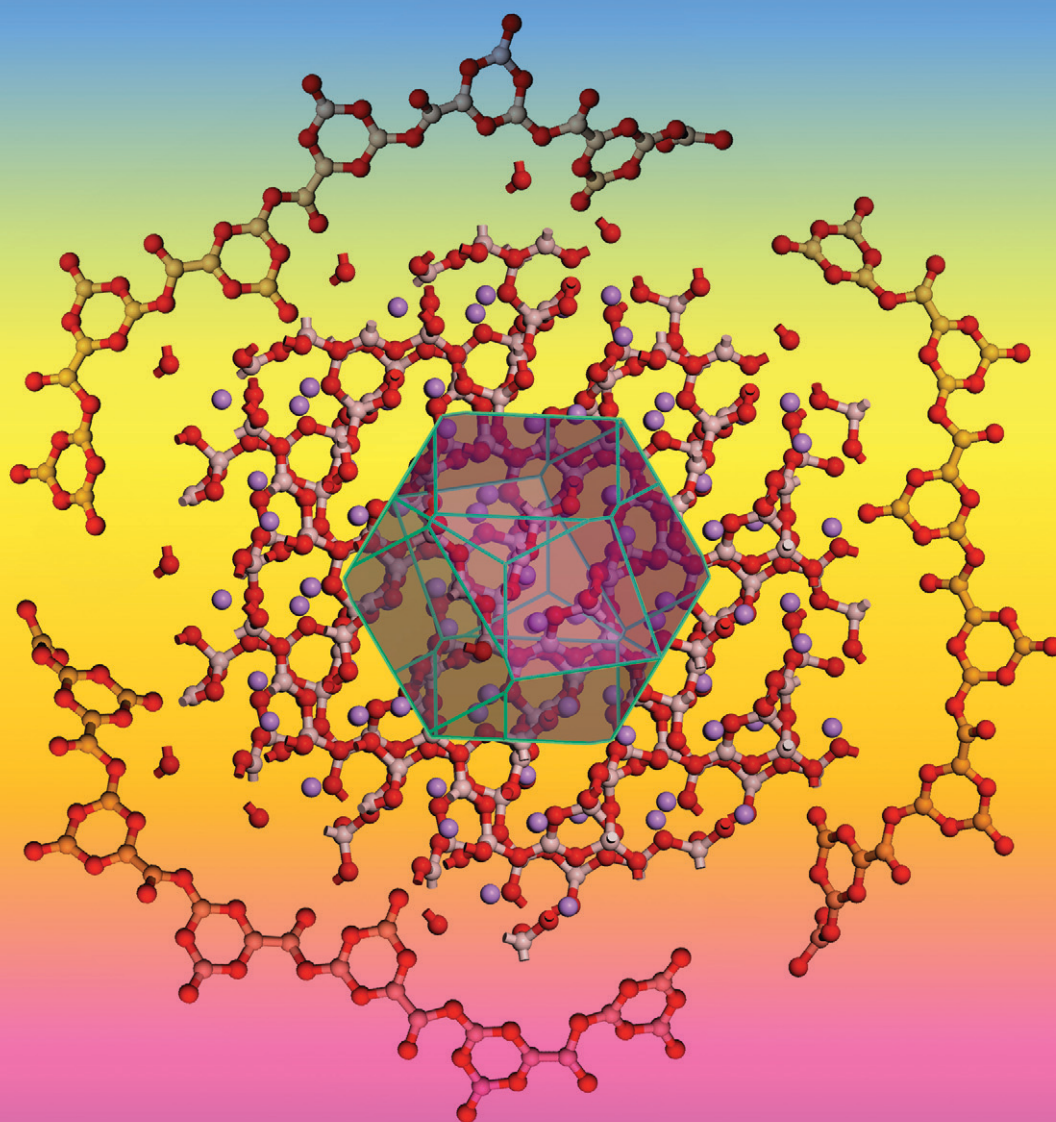


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Raman spectroscopy and density functional theory analyses of the melt structure in a $\text{Li}_2\text{B}_4\text{O}_7$ crystal growth system†

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Melt structure, a fundamental and challenging subject for borate crystal growth, has not been solved for many years. In this paper, a new method has been employed to study the $\text{Li}_2\text{B}_4\text{O}_7$ melt structure. High-temperature Raman spectroscopy has been used to investigate the structural evolution from a $\text{Li}_2\text{B}_4\text{O}_7$ crystal to a $\text{Li}_2\text{B}_4\text{O}_7$ melt. Based on the investigation, a model was proposed to describe the $\text{Li}_2\text{B}_4\text{O}_7$ melt. The melt is made up of polymer-like boron–oxygen chains; the minimal repeated unit is the $\text{B}_4\text{O}_6\text{O}_2^{2-}$ (O = bridging oxygen) group which is formed by a $\text{B}_3\text{O}_4\text{O}_2^-$ six-membered ring and a BO_2^- triangle linked by a bridging oxygen atom. DFT calculations have verified the melt structure and provided accurate assignments for the vibrational bands present in the $\text{Li}_2\text{B}_4\text{O}_7$ melt Raman spectrum.

1. Introduction

Lithium tetraborate ($\text{Li}_2\text{B}_4\text{O}_7$) single crystals have attracted great attention for use in many fields including surface acoustic wave,¹ bulk acoustic wave,² non-linear optics³ and neutron detection.⁴ The crystal is usually grown from a $\text{Li}_2\text{B}_4\text{O}_7$ melt by the Czochralski⁵ or Bridgman method.⁶ High viscosity of the $\text{Li}_2\text{B}_4\text{O}_7$ melt, like other borate melts, is not beneficial to the successful preparation of large and high-quality $\text{Li}_2\text{B}_4\text{O}_7$ single crystals. For example, as a result of the high viscosity, bubbles and voids were often found heavily concentrated at the central part of the crystal boules.^{5,6}

The melt structure is an intrinsic factor governing a variety of melt macro-properties, including viscosity, density and surface tension.⁷ Deep understanding of the $\text{Li}_2\text{B}_4\text{O}_7$ melt structure in a $\text{Li}_2\text{B}_4\text{O}_7$ crystal growth system can help us to better interpret the crystal growth phenomena and then to improve the crystal quality. On the other hand, it is known that the $\text{Li}_2\text{B}_4\text{O}_7$ crystal structure is transformed from the melt structure.⁸ Therefore, some important information on the $\text{Li}_2\text{B}_4\text{O}_7$ crystal growth process on the molecular scale can be obtained based on the

study of the $\text{Li}_2\text{B}_4\text{O}_7$ melt structure. However, despite the importance of the borate melt structure, a detailed study on it has not been conducted up to now. In the past, a borate melt structure was usually extrapolated from the glass structure of the same stoichiometry. Actually, a glass structure is the super-cooled melt structure frozen at the glass transition temperature rather than the true high-temperature melt structure.⁹

Accurate investigation of a melt structure requires high-temperature *in situ* experimental techniques. In comparison to other experimental techniques, Raman spectroscopy is regarded as an effective and convenient *in situ* method for the study of melt structures.^{8,10} However, the conversion of the vibrational peaks seen in a Raman spectrum into the structural information remains difficult.¹¹

Recently, the density functional theory (DFT) method has been used to predict and interpret Raman spectra for a wide range of materials, including crystals, polymers and molecules.¹² It provides a possibility to study a melt structure and simulate its Raman spectrum. In this paper, we proposed a new and general method, which combines high-temperature Raman spectroscopy with DFT calculation, to study the melt structure in the $\text{Li}_2\text{B}_4\text{O}_7$ crystal growth system.

2. Experimental

A homemade micro-furnace used in this work was described elsewhere.⁸ A $\text{Li}_2\text{B}_4\text{O}_7$ crystal slice with a size of $10 \times 5 \times 1 \text{ mm}^3$ was placed into a platinum boat, and then into the micro-furnace with the boat. The micro-furnace provided a horizontal temperature gradient in the boat. The crystal and melt Raman spectra were recorded using a Jobin Yvon LabRaman HR800

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† Electronic supplementary information (ESI) available: Calculated optical modes of the $\text{Li}_2\text{B}_4\text{O}_7$ crystal and melt and their frequencies, atomic configuration of the (112) face in the $\text{Li}_2\text{B}_4\text{O}_7$ crystal structure and the optimized melt structure. See DOI: 10.1039/c3ce42544d

Raman spectrometer in a back scattering configuration. The Raman scattering light was collected by a confocal lens system and then recorded using an intensive charge couple device (ICCD). The excitation source was the 355 nm line of a Q-switch pulsed THG-Nd:YAG laser with an average output of 0.9 W. The diameter of the laser beam was less than 2 μm . The instrumental resolution was $\pm 0.1 \text{ cm}^{-1}$ with a repeatability of $\pm 1 \text{ cm}^{-1}$. The spectral acquisition, under an accumulated mode, was preformed for 50 s integration time (5 scans, scan time 10 s each). The spectral collecting range was 70–2000 cm^{-1} .

3. Computational details

All calculations were carried out using a plane wave basis-set and norm-conserving pseudopotentials as implemented in the CASTEP (Cambridge Sequential Total Energy Package) code.¹³ Exchange and correlation energies were approximated using the Perdew–Burke–Ernzerhof functional of generalized gradient approximation (GGA-PBE).¹⁴ The plane-wave energy cutoff and the convergence criterion of the self-consistent field (SCF) were 950 eV and 10^{-6} eV per atom, respectively. The Brillouin zone integrations were done over a $2 \times 2 \times 2$ Monkhorst–Pack grid for the crystal and over a $2 \times 4 \times 2$ Monkhorst–Pack grid for the melt. The experimentally determined $\text{Li}_2\text{B}_4\text{O}_7$ crystal structure¹⁵ and the proposed $\text{Li}_2\text{B}_4\text{O}_7$ melt structure were used as the initial structures for structural optimization and Raman spectrum calculation.¹⁵ All calculated Raman intensities were corrected by Bose–Einstein factors with the experimental temperatures (300 K for the crystal and 1200 K for the melt) and the excitation source wavelength (355 nm). The Raman peaks were broadened with Gaussian shape functions using the SWizard program.¹⁶

4. Results and discussion

Crystal Raman spectra were recorded at different temperatures. The cases at 30 °C and 500 °C are shown in Fig. 1a and b, respectively. With an increase of temperature, all Raman peaks

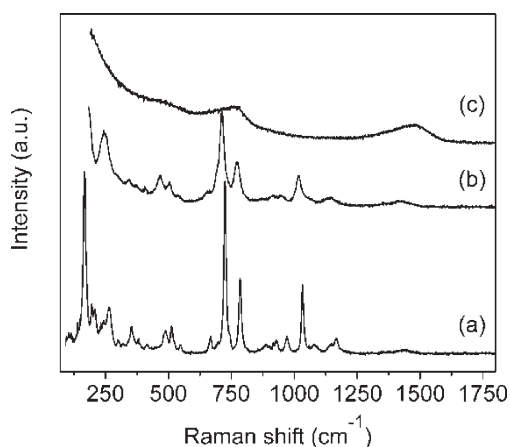


Fig. 1 Raman spectra of $\text{Li}_2\text{B}_4\text{O}_7$ crystal and melt. (a) Crystal Raman spectrum at 30 °C; (b) crystal Raman spectrum at 500 °C; (c) melt Raman spectrum near the melting point.

red-shifted and broadened in different degrees. When the hot end was heated over the melting point, the crystal began to melt. After that, the temperature was slowly decreased until a steady crystal–melt interface was formed (see Fig. 2). The melt Raman spectra were recorded near the interface to ensure the measuring temperature close to the crystal growth temperature. A typical melt Raman spectrum is shown in Fig. 1c. Comparing with the crystal Raman spectra, we found that: (1) the intensive crystal peaks located around 720 cm^{-1} and in the region of 900–1200 cm^{-1} disappeared in the melt Raman spectrum; (2) the Raman peaks in the high frequency region (1300–1500 cm^{-1}) anomalously blue-shifted and increased in intensity when the crystal melted.

Our aim is to obtain the $\text{Li}_2\text{B}_4\text{O}_7$ melt structure in the $\text{Li}_2\text{B}_4\text{O}_7$ crystal growth system. The $\text{Li}_2\text{B}_4\text{O}_7$ crystal structure has been determined by the X-ray single crystal diffraction method. If the structural evolution details from the $\text{Li}_2\text{B}_4\text{O}_7$ crystal to melt are disclosed, the melt structure will be determined. In this work, the evolution details are reflected by the Raman spectrum changes during the $\text{Li}_2\text{B}_4\text{O}_7$ crystal melting process.

In order to determine the structural origins of the spectral changes, some important crystal Raman peaks should be assigned. The $\text{Li}_2\text{B}_4\text{O}_7$ crystal structure solely consists of $\text{B}_4\text{O}_5\text{O}_4^{2-}$ groups (O = bridging oxygen, as shown in Fig. S1†) except for Li^+ ions. Structural complexity of the $\text{B}_4\text{O}_5\text{O}_4^{2-}$ group makes it very difficult to give an exact assignment for each crystal Raman peak. Generally, the Raman peaks were approximately assigned to the vibrations of BO_4^- tetrahedra and of BO_3 triangles (two basic structural units constituting the $\text{B}_4\text{O}_5\text{O}_4^{2-}$ group) although the two units are not independent.¹⁷ In this paper, we used the DFT method implemented in the CASTEP code to calculate the $\text{Li}_2\text{B}_4\text{O}_7$ crystal Raman spectrum for the first time which is intended to give clearer assignments to some important vibrational peaks. The calculated results give 153 optical modes ($18A_1 + 19A_2 + 19B_1 + 19B_2 + 39E$), among these, 134 modes ($18A_1 + 19B_1 + 19B_2 + 39E$) are Raman active (see the ESI,† Table S1). These results are in agreement with group-theory analysis.¹⁷ The calculated Raman spectrum is shown in Fig. 3, along with the experimental spectrum. All calculated frequencies coincide well with the experimental within an acceptable error margin (20 cm^{-1}). The relative intensities of the calculated peaks are

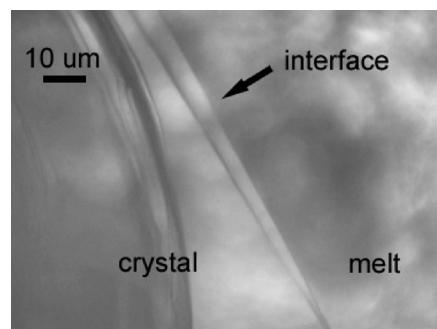


Fig. 2 A typical $\text{Li}_2\text{B}_4\text{O}_7$ crystal–melt interface.

also consistent with the experimental results except for some weak experimental peaks, such as 259 cm^{-1} , 960 cm^{-1} and 1150 cm^{-1} peaks. The theoretical agreement with experiment confirms the validity of our calculation method.

According to the calculated results, some key Raman peaks were assigned. The intensive peaks located in the range of $900\text{--}1200\text{ cm}^{-1}$ mainly arise from the asymmetrical vibrations of the BO_4^- units. The atomic displacements of 955 cm^{-1} and 1021 cm^{-1} peaks are shown in Fig. 4. These peaks disappeared in the melt spectrum, suggesting that the BO_4^- units were converted into other boron–oxygen units during the melting process, such as BO_2^- (a boron–oxygen triangle consisting of two bridging oxygen atoms and one non-bridging oxygen atom) caused by the break of $\text{B}_4\text{--O}$ bonds (the B--O bond of a BO_4^- tetrahedron), as found in many other borate melts.¹⁸ The weakness of the $\text{B}_4\text{--O}$ bonds is also supported by the crystal habit. For example, Pan *et al.* found that the $\text{Li}_2\text{B}_4\text{O}_7$ crystal easily cracks along the $\{112\}$ faces which are connected by the $\text{B}_4\text{--O}$ bonds (see the ESI,† Fig. S1).¹⁹

The vibrational bands in the $600\text{--}850\text{ cm}^{-1}$ region are related to the breathing vibrations of boron–oxygen six-membered rings.^{18,20} During the melting process, the crystal 720 cm^{-1} peak disappeared and the 770 cm^{-1} peak remained, indicating boron–oxygen six-membered rings still remained in the melt although their structures were different from those in the $\text{Li}_2\text{B}_4\text{O}_7$ crystal structure.

The high-frequency crystal peaks located around 1435 cm^{-1} are related to the stretching vibrations of the extra-ring $\text{B}_3\text{--O}$ bonds (the B--O bond of a BO_3 triangle), as shown in Fig. 4c. The peaks anomalously blue-shifted to 1480 cm^{-1} in the melt Raman spectrum, indicating new boron–oxygen structural units formed in the melt. Generally, strong Raman peaks located around 1500 cm^{-1} are attributed to the stretching vibrations of extra-ring $\text{B}_3\text{--O}$ bonds,²⁰ which implies that the extra-ring $\text{B}_3\text{--O}$ bonds in the crystal structure changed to the extra-ring $\text{B}_3\text{--O}$ bonds when the crystal melted.

By investigating the structural evolution in the melting process, we found that the $\text{B}_4\text{--O}$ bonds in the $\text{Li}_2\text{B}_4\text{O}_7$ crystal structure were broken. As a result, new boron–oxygen six-membered

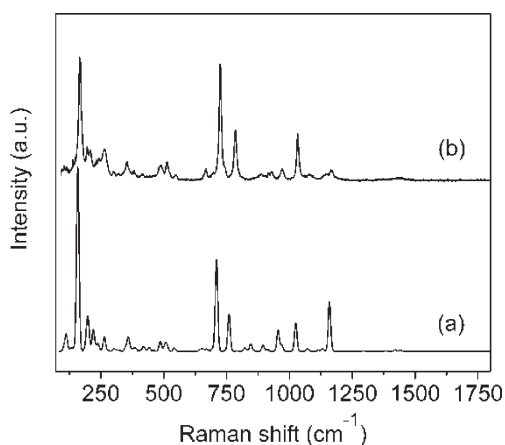


Fig. 3 (a) Calculated and (b) experimental Raman spectra of $\text{Li}_2\text{B}_4\text{O}_7$ crystal.

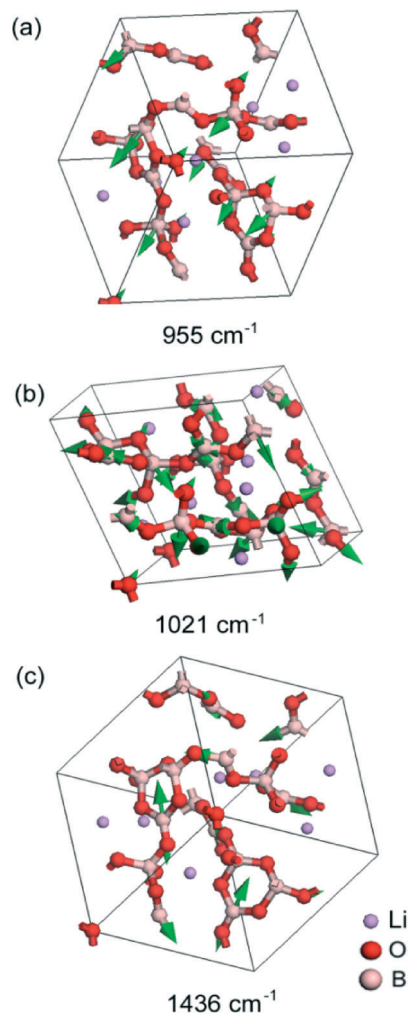


Fig. 4 Atomic displacements of three important crystal vibrational peaks.

rings with extra-ring non-bridging oxygen atoms appeared in the melt. In terms of the structural evolution, we deduced the $\text{Li}_2\text{B}_4\text{O}_7$ melt structure. The melt is made up of polymer-like boron–oxygen chains; the minimal repeated unit of the chain is the $\text{B}_4\text{O}_6\text{O}_2^{2-}$ group which is formed by a $\text{B}_3\text{O}_4\text{O}_2^-$ six-membered ring and a BO_2^- triangle linked by a bridging oxygen atom, as shown in Fig. 5b.

The $\text{Li}_2\text{B}_4\text{O}_7$ melt structure was verified by DFT calculation. In general, DFT calculations are performed in three-dimensional periodic systems; however, the $\text{Li}_2\text{B}_4\text{O}_7$ melt is non-periodic. In order to simulate the melt structure, a super-cell method was used.²¹ One minimal repeated unit ($\text{B}_4\text{O}_6\text{O}_2^{2-}$ group) was placed into a unit cell; two Li^+ ions were put into the same unit cell to balance the charge of the boron–oxygen group (see Fig. 5b). Considering the strong electrostatic interactions between the chains and Li^+ ions, we did not limit the unit cell parameters and the atomic positions during the DFT geometry optimization.²² After the most stable structure was achieved, its Raman spectrum was calculated at the same level of theory for the $\text{Li}_2\text{B}_4\text{O}_7$ crystal Raman spectrum calculation.

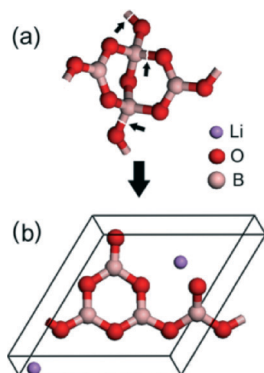


Fig. 5 Structural evolution of the $\text{Li}_2\text{B}_4\text{O}_7$ crystal to the $\text{Li}_2\text{B}_4\text{O}_7$ melt. (a) Basic boron-oxygen structural group in the $\text{Li}_2\text{B}_4\text{O}_7$ crystal; (b) basic structural group in the $\text{Li}_2\text{B}_4\text{O}_7$ melt.

The optimized melt structural unit (the minimal repeated unit) is shown in Fig. 5b. The calculated B–O bond lengths and O–B–O angles are all in agreement with the reported values (see the ESI,† Fig. S2 and Table S2). The calculated results give 36 modes (36A) of Raman active, which is in agreement with that of group-theory analysis (see the ESI,† Table S3). The calculated $\text{Li}_2\text{B}_4\text{O}_7$ melt Raman spectra, together with the experimental spectrum, are shown in Fig. 6. The temperature dependence (thermal broadening) of the Raman peak shape was simulated by a Gaussian line shape function with a FWHM (full width at half maximum) of 200 cm^{-1} .¹⁶ The superposition of these Gaussian profiles is shown in Fig. 6b. All of the calculated frequencies and intensity are consistent with the experimental results, which confirms our inference that the $\text{Li}_2\text{B}_4\text{O}_7$ melt is made up of the $\text{B}_4\text{O}_6\text{O}_2^{2-}$ chains.

The calculated results further provide accurate assignments for the three main vibrational bands present in the melt Raman spectrum (see Fig. 6). (1) The Raman band in the range of $250\text{--}500\text{ cm}^{-1}$ is related to the motions of the

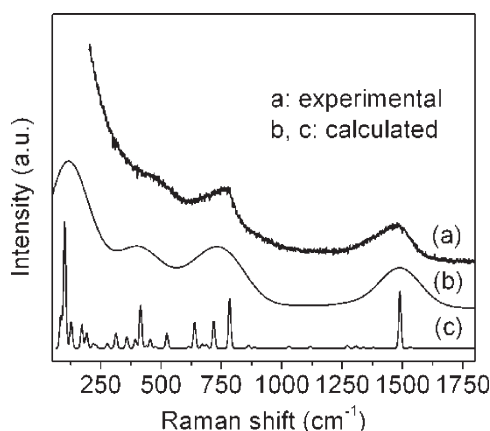


Fig. 6 Experimental and calculated Raman spectra of the $\text{Li}_2\text{B}_4\text{O}_7$ melt. (a) Experimental Raman spectrum; (b) calculated Raman spectrum broadened by the Gaussian line shape function with a FWHM of 200 cm^{-1} ; (c) calculated Raman spectrum broadened by Gaussian line shape function with a FWHM of 10 cm^{-1} .

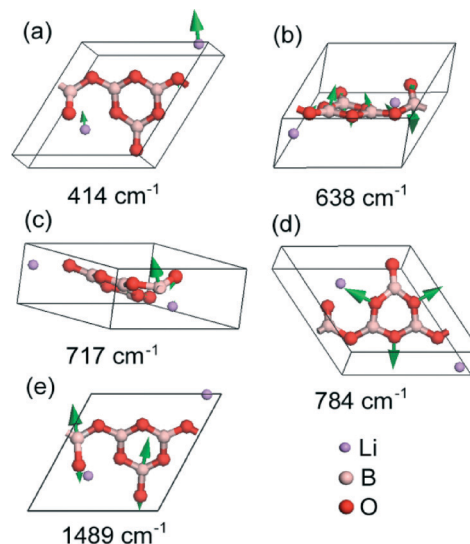


Fig. 7 Atomic displacements of five strong peaks in the calculated $\text{Li}_2\text{B}_4\text{O}_7$ melt Raman spectrum.

Li^+ ions. (2) The Raman band in the range of $600\text{--}800\text{ cm}^{-1}$ includes three strong peaks, respectively located at 638 cm^{-1} , 717 cm^{-1} and 784 cm^{-1} . The 638 cm^{-1} peak and the 717 cm^{-1} peak are mainly attributed to the out-plane bending vibrations of the boron-oxygen triangle; the 784 cm^{-1} peak is related to the breathing vibration of the boron-oxygen six-membered ring. (3) The high-frequency band from 1300 cm^{-1} to 1600 cm^{-1} primarily arises from the stretching vibration of the extra-ring $\text{B}_3\text{--O}$ bond. The atomic displacements of the five strong peaks are shown in Fig. 7.

5. Conclusions

In order to obtain the melt structure in the $\text{Li}_2\text{B}_4\text{O}_7$ crystal growth system, Raman spectroscopy and DFT methods were used to investigate the structural evolution from $\text{Li}_2\text{B}_4\text{O}_7$ crystal to melt. During the crystal melting process, the $\text{B}_4\text{--O}$ bonds in the crystal structure were broken. As a result, new boron-oxygen six-membered rings with extra-ring non-bridging oxygen atoms appeared in the melt. According to the structural evolution, we proposed a structural model to describe the melt. The melt is made up of polymer-like chains; their minimal repeated unit is the $\text{B}_4\text{O}_6\text{O}_2^{2-}$ group which is formed by a $\text{B}_3\text{O}_4\text{O}_2^-$ six-membered ring and a BO_2^- triangle linked by a bridging oxygen atom.

The Raman spectrum of the $\text{Li}_2\text{B}_4\text{O}_7$ melt was calculated by the DFT method. The calculated Raman spectrum is consistent with the experimental. The Raman band in the range of $250\text{--}500\text{ cm}^{-1}$ is related to the motions of the Li^+ ions; the Raman band in the range of $600\text{--}800\text{ cm}^{-1}$ is mainly attributed to the out-plane bending vibrations of the boron-oxygen triangle and the breathing vibration of the six-membered ring; the Raman band in the range of $1300\text{--}1600\text{ cm}^{-1}$ primarily arises from the stretching vibrations of the extra-ring $\text{B}_3\text{--O}$ bond.

We believe that the method provided in this paper can be extended to investigation of other borate melt structures and help us deeply understand the macro-properties of borate melts and the micro-process in borate crystal growth.

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